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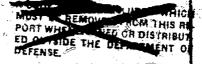
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288 106



Rock Island Arsenal Laboratory



TECHNICAL REPORT

DEVELOPMENT OF FLEXIBLE INSULATION FOR SOLID PROPELLANT ROCKET MOTOR CASES

Ву

R. E. Ofner

and

D. H. Sale

Department of the Army Project No. 593-32-008

Ordnance Management Structure Code No. 5010.11.843

Report No. 62-2366

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Rock Island Arsenal Rock Island, Illinois

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ABSTRACT

The development of flexible, solid propellant rocket motor case insulation is discussed. Data are presented for insulation based primarily on butadiene/styrene and butadiene/acrylonitrile copolymers. The oxyacetylene torch test currently being standardized by the Flame Ablation Test Group of Section III-L of ASTM Committee D-20 was the principle screening tool used in the study.

Vulcanizates were compounded using a variety of fillers and filler combinations including salts, resins and fibers. The selection of these fillers was based on such properties as heat stability, ability to form char, heat capacity and their known ability to reinforce rubber. As a result of these studies, a material was developed which static motor tests show to have promise as a flexible material for case insulation. This material was based on a butadiene/acrylonitrile-phenol furfural-asbestos composition.

Other promising materials which are reported are based on two types of liquid butadiene/styrene copolymers and a butadiene/acrylonitrile-polyvinyl chloride blend.

RECOMMENDATIONS

It is recommended that a double bladed, Sigma type mixer be obtained in order that more uniform mixes of liquid polymers and fillers can be obtained without destroying the integrity of the filler itself.

It is recommended that commercially available liquid polymers be evaluated with long fiber asbestos as a filler in order to determine the superiority of one polymer over another.

The best insulation material developed to date in this study is a butadiene/acrylonitrile-phenol furfural-asbestos vulcanizate. It is recommended that a study be conducted to optimize this composition as to quantity and types of ingredients; for example, it might be possible to substitute a liquid polymer in place of the solid polymer.

Despite the poor elongation of resin filled insulation materials, the good insulating ability and erosion resistance of these vulcanizates warrant their further study. It is recommended that special efforts be made to evaluate fully the flexible resins developed by Atlantic Research Corporation under Contract #DA-036-ORD-3325RD as well as commercially available flexible resins.

DEVELOPMENT OF FLEXIBLE INSULATION FOR SOLID PROPELIANT ROCKET MOTOR CASES

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DEVELOPMENT OF FLEXIBLE INSULATION FOR SOLID PROPELIANT ROCKET MOTOR CASES

OBJECT

To develop improved, flexible, thermal case insulation for solid propellant rocket motors.

INTRODUCTION

Lighter weight structural materials and more efficient propellants are among the major factors which are expected to improve the performance of solid propellant rocket motors. Concomitant with the use of these materials is the need for more efficient rocket motor insulation. The lightweight metallic and plastic case materials generally suffer from low tolerance to heat. The super propellants burn for several minutes at very high temperatures and produce high pressures and erosive combustion products. These factors all indicate the need for improved insulation. Superior insulation is especially needed for rocket motors which employ end-burning grains. Batchelor et al. point out that "the capability of rockets using end-burning grains can be limited by the motor case insulation because this material must survive exposure to the hot gases for the total duration of the firing."

Case insulation must possess the obvious characteristics of heat and erosion resistance in order to withstand the severe operating environments but in addition must exhibit the less obvious property of flexibility. The need for flexible case insulation has been cited by only a few workers in the open literature. Batchelor et al (1) state that "the liner [insulation] need not add mechanical strength to the case, but it must maintain its integrity while being strained to conform to the case deformation which occurs at initial pressurization and may reach 0.8 percent." Batchelor et al(2) further state that "the selection of rubber binders is often prompted by a realization that flexibility and reasonable elongation in a motor case insulation is generally desirable and often imperative." Shapiro and Hughes (3) describe rubber based insulation which has an ultimate elongation of 600 percent. This very high elongation was not the goal of these workers, it was merely the value obtained for the insulation which they had developed.

What degree of flexibility 's required in case insulation? It would appear that the state of the art has not advanced sufficiently to provide answers to this question.

It is the opinion of the authors that Batchelor's value of 0.8 percent elongation is too low and does not provide a margin of safety. The 600 percent value of Shapire and Hughes seems to be unrealistically high.

It has not been the purpose of the present study to determine the optimum degree of flexibility for case insulation nor has the work been pointed toward any particular rocket motor system. The primary aim has been the development of insulation materials having maximum flexibility and minimum density, with both of these properties mederated to provide insulations consistent with the fundamental requirements, namely, resistance to flame and erosion.

The many desirable features of case insulation, viz., compatibility with and bondability to grain and case, ease of manufacture from commercially available materials and high degree of reliability during long term storage have been kept in mind but have not been the main guidelines for the development work reported herein.

A major difference exists between the work reported herein and the earlier work (7) in that the test method used here is much more severe.

PROCEDURE

Screening of candidate insulation materials was performed with an exyscetylene torch test. A schematic diagram of the test equipment is shown in Figure 1 and test conditions are presented in Table 1. The equipment and precedures duplicate those of the test currently being standardized by the Flame Ablation Test Group of Section III-L of ASTM Committee D-20. Further details pertinent to this type of test have been published (4) by the Naval Ordnance Laboratory, the agency largely responsible for the current efforts to standardize the torch test.

The effectiveness of candidate insulation materials was measured by two test criteria. One was the rate of temperature rise on the back side of the specimen while the front side was exposed to the expacetylene terch flame. The other was the time required for the flame to burn through the specimen. The results of the tests obtained in such manner are reported as a performance index and an erosion rate. The index, referred to as P200, is computed by dividing the time (seconds) required for the specimen back side to reach 200°C by the original specimen thickness (centimeters) and by the specific gravity. The erosion rate, E, is computed by dividing the original specimen thickness (mils) by the burn through time in seconds. It should be noted that high

SCHRMATIC OF RIA OXYACETYLENE TORCH APPARATUS

TABLE I

TORCH TEST OPERATING CONDITIONS

Oxygen flow rate, standard cubic fee	t/hour (SCFE	1), 127
Acetylene flow rate, SCFH,		97
Volume ratio of oxygen to acetylene,		1.3
Impingement angle between flame and specimen, degrees,	*.	90
Specimen size, inches,	4. **	4 x 4 x 1/4
Distance from torch tip to specimen,	inches,	3/4
Temperature of oxygen and acetylene,	oc.,	24±3
Method of determining moment of burn	through,	visual

values of P200 and low values for E are indicative of good insulation properties. Unless otherwise noted, the performance indices and ercsion rates reported in this work are the average of two tests. The average variance of the performance index is ± 3 points, that of the erosion rate about ± 0.2 points.

Those materials which exhibited excellent thermal insulation capabilities during the screening test were further evaluated in static rocket motor firing tests conducted by the Allegany Ballistics Laboratory (ABL) and the Atlantic Research Corporation (ARC) at their respective test facilities. Details of the static firing test methods have been published in a classified report (5).

Formulations and curing conditions for all compositions tested are given in Table II. Test specimens were molded in a four cavity mold.

Unless otherwise noted, all compounds were mixed, cured and tested for stress-strain properties in accordance with the applicable ASTM(6) procedures.

RESULTS

One of the major findings of the earlier work (7) on case insulation conducted at the Rock Island Arsenal Laboratory was that the behavior of unfilled (gum) rubber vulcanizates, when tested in an oxyacetylene flame, depended upon the type of polymer present in the vulcanizate. This observation was based on data obtained with an exyacetylene torch test similar in most respects to the ASTM proposed standard test described in Figure 1 of this report but utilizing a lower velocity flame which burned at a cooler temperature. effort to determine whether the higher velocity, hotter flame was also capable of differentiating among the insulation abilities of various gum vulcanizates, a portion of the earlier work on gum vulcanizates was repeated with the ASTM proposed test. The results obtained with the two torch tests and the major respects in which the tests differ, are given in Table III.

It is readily apparent from the data in Table III that the low velocity torch test provides discrimination among gum vulcanizates based on different polymers whereas the high velocity test prevides no discrimination. All specimens tested in the low velocity flame charred to varying degrees. It is believed that the amount, type and strength of the char determined the length of time required for heat to ponetrate the specimens. In the higher velocity test, however, no visible char was formed on any of the specimens.

TABLE II

COMPOUND FORMULATIONS

					24	Parts By Weight	1ght				
	1			,	raule Nu	mbers for	Formula Numbers for Basic Series	- F-18			
Compounding ingredients	HOLM	N 39	NA.	N141C	M41C3	N141C3D3	N142C3D4	N141C3D5	270	Z56C5	2101
Polychloroprene 55/45 Butadiene/acrylonitrile	100	100	700	700	100	•	-	Э.			
80/20 " "					, ,	100	~ ;	81			
Entadiene/acrylonitrile-						÷	001				-
polyvinyl chloride blend	2	· .						. 001	٠.		-
Methyl phenyl vinvl #11cone	٠, .								100		
Vinyliders fluoride/hexa-	e	-			. 47				- 30	700	
fluoropropylene					-						9
Stearic acid	ւ.	1,5	~		8	8	8				707
Aine Oride	io e	rO.	·0		n	· 67	ım	i m	:		
Phenyl-bata-naphthylamine	k 64						, ¹		- ,		15
Stilfur	ļ.	~			64	er.	b	٠			
Serrothiaryl disulfide		-			7.5	2.5	, r	۲. ا			
Juriametrical Di-bate-markthylls Symmetrical Di-bate-markthyls	•	īĞ.	m		81.	77.	113	.12			
phenyleuediamine		(r)		·							
Di-tert-butyl peroxide		,	3							-	
Dicumyl percentde (40% active)									4	ic)	
	-					٠	,			1	83

fallers as indicated in applicable tables.

Compound Z40 press cured 30 minutes #1539C, post cured in an air oven 6 hours #232°C. Compound Z56C5 press cured 10 minutes #135°C, post cured in an air oven 8 hours #136°C. Compound Z101 press cured 35 minutes #138°C, step cured in an air oven 1 hour #100, 121 post cured 24 hours #204°C.

All other compounds press cured 35 minutes #163°C.

-=:			rormaly	Numberrs 1	Formula Numbers for Basic Series	Series		
			S77CID		-			
Compounding Ingredients	211	1022 T	S77C1D4	S77CIDI	STTCIDS	87703	S77C4D2	\$77C4D3
76.5/23.5 Butadiene/styrene 60/40 Butadiene/styrene Cis 1,4 Polyhutadiene	700	000	100	100	, 001	300		. e :
High Viscosity Liquid Butadiene/ styrene			*,			2	100	
styrene	f					,		100
Stearic acid	RI M	74 M	91 PA	ณ ๓	64 M	લ ભ		
Sulfur	1.75	m	, ,	m	m	1.5	77	7
Symmetrical Di-bets-naphthyl-p-	i	i	,	,	,			
phenylenedizmine Benzothiszyl dizulfide	ri.	8 H F	1 7	7	7	. rd	4	•
Tetramethyl thiuram disulfide		112	.12	.12	.12	ļ	l	
Zinc disthyldithiocarbamate							1.1	1.1
_							•	

TABLE III

THE EFFECT OF FLAME TEMPERATURE AND GAS VELOCITY ON THE INSULATING ABILITY OF GUM VUICANIZATES

**	,-:,-: -		Low Velocity Test	y Test	AST	ASTM Proposed High Velocity Test	High
	App	pproximate flame temp., oc.	2900			3300	
	Total	al gas flow rate, SCFH	34		-	224	
A S S	RIA Formula No.	Polymer Type T	Time to Reach Backside* Temperature of 200°C, Sec.	ackside*	Temperat	ure of 200	OC, Sec.
2		Vinylidene fluoride/ hexafluoropropylene	120			•	
=	10EF1	Polychloroprene	48		v 	2	
Z	68 139	55/45 Butadiene/acrylo- nitrile	1.7			6	
Š	STTF	76.5/23.5 Butadiene/ styrene		÷ .		E.	
	*A11 two	specimens were 0.250 ± 0.005" thick. All tests. Tests were reproducible to within	.005" thick.		results are 15 percent.	the average of	e of

It is assumed that if char was formed, it was immediately blown away by the high velocity flame. Thus, in the newer test, all specimens were penetrated by the heat of the flame very rapidly and at equal rates.

Many investigators have reported that the efficacy of case insulation materials depends to a large degree upon the ability of the predominant polymeric ingredient to form low molecular weight gases. The gases add to the insulation efficiency by abscrbing heat while flowing through the charred layer, through the process termed "transpirational cooling". No investigator, however, has formed definite conclusions as to which polymers provide optimum transpirational cooling when insulation is burned under the conditions of the torch test or those of actual use. Lacking this information. and realizing that the torch test was of no value in discriminating among polymers on the basis of tests on gum vulcanizates, the selection of the polymers for use in a study of filled vulcanizates became somewhat arbitrary. The two polymer types chosen for the major portion of this study, butadiene/styrene (SBR) and butadiene/acrylonitrile (NBR). were selected because of their current use in commercial insulations, their compatibility with a wide variety of fillers, low densities and low cost.

The torch evaluation results for vulcanizates based on SBR and containing fillers are given in Table IV. The data are arranged into groups, according to the types of fillers used. Choice of fillers was made on the basis of inherent heat resistance, ability to reinforce rubber, capacity to absorb heat during change in state or ability to form highly crosslinked systems.

Before analyzing the data of this report, it is important that the reader understand the significance of the performance index and the erosion rate and to be aware of the relationship between these test criteria. It should be apparent that a material which is a good thermal insulator will require a long period of time to attain a backside temperature of 2000c, and it will, therefore, show a high index. Its erosion rate will usually be low but the relationship between P200 and E will not necessarily be proportional, because the two values are determined at different points in time. Two materials having equal P200 values may exhibit grossly different E rates, depending upon their performance after the 200°C. temperature is reached. For example, one material may have a high coefficient of thermal conductivity but may be very resistant to flame penetration and eresien, in which case it will exhibit a poor (low) P200 but a good (low) E rate. This situation is best exemplified

TABLE IV

TORCH TEST DATA FOR FILLED SER VULCANIZATES

FIBROUS FILLERS**

Tensile Slong. Strength % psi	55 1050	95 425	1	40 190	135 460		80 800	00 920		235 170				
R. El	5***	~	13	10 17	15	15 3	14	15 2	18 . 4	22	28	25	41 4	ber.
P200	21***	38	28	26	21	19	18	18	14	12	12	11	∞	s of rub
Amownt,	100	20	_	100	20			100			20		20	100 part
	tos	toes			108	-	COB			ø				H
Type of Filler	Chrysotile long fiber asbestos	Chrysotile long fiber asbes	Acrylic fiber	Chopped glass	Chrysotile med. fiber asbest	Ceramic fiber	Chrysotile med. fiber asbesi	Potassium titanate	Chrysotile asbestos floats	Long staple aluminum silicat	Chrysotile asbestos floats	Chopped aluminum silicate	Chopped aluminum silicate	represents parts by weight per 100 parts of rubber

**All compounds except those containing acrylic fiber and asbestos floats were mixed with mill roll gear ratio 1:1 rather than the standard (ASTM-D-15) 1.4:1 ratio, in an attempt to preserve fiber length.

***Average of 15 tests.

TABLE IV

RESINOUS PILLERS

Tensile Strength, psi	I :	240	260	1	180	300	150	240	230	180	280	06
Elong.	to brittle	100-200	65	1	120	135	130	65	380	265	305	1270
B.	6	· O:	4	m	11	14	n	24	25	56	28	41
P200	52	44	40	37	32	28	26	14	14	14	12	∞
Amount, PHR*	100	700	TOO	125	20	20	75	100	100	20	25	100
Type of Filler	Phenol formaldehyde #1	Phonol tornatuenyde #2		Phenol furfural	formaldehyde	Phenol formaldehyde #1	Phenol furfural	Phenol formaldehyde	Phenol formaldehyde (oil modified)		Phenol furfural	Phenol formaldehyde #3
RIA Formula No.	S77C1F34	C77C1 E42	C#4170000	511C1F13	S77C1F50	SYZCIF51	S77C1F74	S77C1F31	S77C1F45	S77C1F73	S77CIF112	S77ClF46

*PHR represents parts by weight per 100 parts of rubber.

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TABLE IV (Cont.)

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MISCELLANEOUS FILLERS

Tensile Strength, psi	100	110	099	160	740	140	1700	80	100	610	540	340	440	170
Elong.	370	70	480	470	470	290	275	130	525	775	645	340	420	270
Ħ	7	20	18	5 4	24	24	3 6	27	17	24	5 6	58	30	38
P200	36 27	18	15	13	11	11	11	11	10	10	10	6	6	9
Amount, PHR	100	100	100	100	100	100	100	100	100	200	100	100	100	100
Type of Filler	Pentaerythritol Med. thermal carbon black	Powdered polyamide plastic	Magnesium fluoride	Powdered sugar	Potassium silicate	Sodium silicofluoride	Med. thermal carbon black	Potassium oxalate	Powdered copper	Magnesium silicate	Wagnestum silicate	Boron carbide	Lithium fluoride	Powdered molybdenum
RIA Formula No.	S77C1F44 S77C1F6	S77C1F39	S77C1F24	S77C1F42	S77C1F29	S77C1F40	S77C1F35	S77C1F25	S77C1F33	S77CIF72	S77C1F71	S77C1F32	S77C1F23	S77C1F41

by the very hard, rigid materials known as carbon-graphites. Typical rubber based insulation containing less than 50 PHR of carbon black or silica fillers will exhibit a poor $^{\rm p}200$ as well as a poor ercsion rate. For the purposes of this study, it was decided that only those insulation materials having indices over forty and erosion rates lower than five were worthy of further evaluation.

Examination of Table IV indicates that of all the vulcanizates in the first group, those containing chrysotile long fiber asbestos showed the best insulation properties. The good flame resistance of these vulcanizates is attributed to the fact that the asbestos filled vulcanizates formed profuse, strong chars which remained attached to the test specimens during the entire test, thereby insulating the substrate material against the heat of the torch. Vulcanizates containing shorter fiber length asbestos also formed large amounts of char but these chars were weak and spalled off under the impact of the high velocity gases. It is interesting to note the tensile strength values of the fiber-containing vulcanizates. The tensile value of the compound containing 100 PHR long fiber asbestos is the highest of the group, indicating that perhaps there is some correlation between tensile strength and ability to form strong char. The correlation is far from absolute, as evidenced by the fairly high tensile strength but poor performance imparted by potassium titanate. It is important at this point to note the useful temperature limits for chrysotile asbestos, ceramic and aluminum silicate fibers. The values are 815, 1140 and 1250°C., respectively. Thus, the least heat resistant fiber produces the best insulation, when all three are compared in SBR vulcanizates at equal weights. This information lends some credence to the belief that the superiority of asbestos as a filler in rubber-based insulation is due in part to its inherent strength. Those vulcanizates which contain brittle fibers such as chopped glass, ceramic or aluminum silicate, all had low tensile strengths, probably because of the cutting action of the fillers when under tension.

In the second group of fillers in Table IV, it is noted that only three of the six phenolic resins which were evaluated showed any promise as fillers for SBR based insulation. The vulcanizates containing 100 PHR of the phenol formaldehyde resins 1 and 2* and the phenol furfural resin were the only ones to exhibit good torch performance. However, the vulcanizate containing 100 PHR resin #1 was very brittle and this resin is, therefore, considered unsatisfactory for use with SBR. Poor dispersion of resin is indicated by the large

^{*}For trade names see Code Sheet at end of report.

variation in ultimate elongation in the case of the phenol formaldehyde resin #2. The phenol furfural resin appears to be the most suitable of this group of fillers for use with SBR.

None of the exides showed promise as fillers for SBR.

Most of the vulcanizates of this group burned with a "popping out" of the filler. The compound containing hydrated silica was the only one to form a char but the char spalled rapidly. Again, as in the case of the fibreus fillers, there appears to be correlation between the ability of a filler to reinforce the rubber and the ability of the rubber to form a char.

Although none of the group of miscellaneous fillers imparted good torch resistance, two are of interest. The low but nonetheless significant effectiveness of pentaerythritol may be caused by its interaction with the rubber. The very low elongation of this vulcanizate indicates a high degree of crosslinking, perhaps caused by the large number of reactive sites and the molecular symmetry of pentaerythritol. The vulcanizate containing 100 PHR of carbon black showed surprising data; high tensile strength but poor torch performance. The explanation to this somewhat anomalous behavior is not known, but may be due to the fact that carbon black burns.

The study of single fillers used in SBR vulcanizates was followed by work on combinations of fillers in this polymer, as reported in Tables V through VII, inclusive.

Table V prevides torsh data for SBR-based compounds containing various combinations of long fiber chrysotile asbestos* and a phenol furfural resin. These fillers had provided good insulation to SBR vulcanizates when used separately, as shown by the data for the two centrol compounds. The use of these fillers in combination provided only marginal improvement in torsh performance. The best compound, centaining 75 PHR of asbestes and 50 PHR of resin, had a higher P200 value then that of the controls but it showed only little improvement in erosion rate.

The last three entries of Table V are performance data for compounds each containing 100 PHR of asbestos and of phenolic resin, but each based upon a different polymer. There is essentially no difference in P200 for these compounds but the erosion rates differ significantly, the compound containing the 60/40 butadiene/styrene polymer having the lowest E rate and the one containing the 76.5/23.5 ratio polymer having the highest rate. These differences may be

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^{*}In the remaining postions of this report the word "chrysotile" will be chitted from "long fiber chrysotile asbestos".

TABLE V

TORCH TEST DATA FOR SBR VUICANIZATES CONTAINING ASBESTOS AND PHENOLIC RESIN

RIA Formula No.	Long Fiber Asbestos	Phenol Furfural	P200	×	Elong.	Remarks
S77C1F22 S77C1F43	100	100	51 40*	10 4	65 55 55	Control Control
S77C1F110	22.55	50	42	9	130	
S77CIF108	25	100	48		្តាល	Elong. measured at 1"/min
S77C1F105	20	20	52	· •	9	
S77C1F113 S77C1F84	50 50	75 100	48 47	ರ ಎ	80 32	Elong. measured at l"/min
S77C1F106 S77C1F114	75 75	25 50	53 59	44	100	
S77C1F77 S77C1F88	100	75 100	46 52**	ည် # #	30	Polymer was Bd/St, 76.5/
S77C1D1F88	100	100	26	₩	20	Polymer was Cis 1,4 poly-
S77CIDF88	100	100	1S	m [']	32	Polymer was Bd/St, 60/40

*Average of four tests.

TABLE VI

TORCH TEST DATA FOR SBR VULCANIZATES CONTAINING CERAMIC FILLERS

	Typ	es and	pes and Amounts (PHR) of Fillers	(PER)	of Fi	llers					
RIA Formula No.	Long Fiber Asbes- tos	A1203	Hy- drated S10 ₂	T102	MgO	Fe203 CaO P200	CaO	P200	E.	Elong.	Remarks
S77C1F78	20	70	56	2.5	0.5	0.5	0.5	40*	*9	190	Sl. spalling.
S77C3F78	20	0.2	56	2.5	0.5	0.5	0.5	25	. 	575	Reduced quantities of rubber curatives.
S77C1F16			100					25	12	325	
S77CIF86		02	78		O. 5	0.5	0.5	17	16	795	Ceramic-type char which spalled quite readily.
S77CIF13		100		9 '.		24		12	21	315	

*Average of six tests.

TABLE VII

THE EFFECT OF POTASSIUM OXALATE ON THE TORCH
PERFORMANCE OF SBR VULCANIZATES

	Elong.	Ł	10 # 10 #	65 250	190 75
	Ä	4 10	0 ro	4 19	9 10
	P200	52 46	47 54	40 I.8	40
llers	Potassium Oxalate	09	09	09	09
(PHR) of F1	Oxide Mixture*	100			100
Types and Amounts (PHR) of Fillers	Long Fiber Asbestos	50	50		50
Types	Phenol Furfural Resin	100	100	100	
	RIA Formula No.	S77C1F80 S77C1F81	S77C1F84 S77C1F83	S77C1F43 S77C1F95	S77C1F78 S77C1F82

*The composition of this oxide mixture is given in Table VI, Formula No. S77CIF86.

^{**}Gas bubbles formed in the vulcanized test pads; stress-strain properties could not be measured.

attributable to differences in rates of gas evolution, types of gases generated from each polymer, the fiber length of the asbestos, or the viscosity of the rubber on the mill affecting fiber length.

Table VI presents terch data for compounds containing ceramic-type fillers. Among these is a compound (S77ClF86) which contains six different metallic oxides in the proportions in which they are commonly used (8) in manufacturing high-alumina fire clay. This compound represented an attempt to produce in situ a high temperature resistant refractory material. The char which resulted from the burning of this compound was, indeed, glassy but it spalled readily and performance of this compound in the flame was poor, being intermediate between a compound containing 100 PHR alumina and one with 100 PHR silica. The high elongation of the compound containing the oxide mixture was encouraging, therefore, asbestos was included with the cxide mixture in an attempt to keep the oxide char from spalling. The resulting material (S77ClF78) did indeed exhibit reduced spalling and improved torch performance. As expected, the addition of asbestos reduced the ultimate elongation. An attempt was made to increase the elongation by reducing the quantity of rubber curatives (\$77C3F78), thus producing an undercure and greater elongation. The expected effect tock place but unfortunately torch performance suffered.

The results in Table VII show the effect on torch performance of the addition of potassium oxalate to SBR vulcanizates. The results show that the addition of the oxalate produced rather marginal improvement in the case of the compounds containing phenolic resin and asbestos or the oxide mixture and asbestos. In the case of the compounds containing resin, asbestos and oxides or resin alone, the addition of potassium oxalate impaired torch performance. Salts such as potassium exalate should be desirable for use in case insulation because of their potential as transpirational cooling agents and because, in general, they have relatively low densities. They have the shortcoming, however, of being hygrescopic. Their affinity for water might adversely affect the processing of compounds containing them and might also adversely affect the storage stability of the fabricated insulation.

Only a limited amount of development work was performed with butadiene/acrylonitrile (NBR) copolymers. The most noteworthy results were obtained with a 55/45 butadiene/acrylonitrile, which was selected because the higher nitrile-containing members of the NBR class are more compatible with phenolic resins. The results of work with this NBR polymer

are given in Table VIII. The first two compounds listed have higher performance values and lower erosion rates than those of any of the other compounds described in this report and are the best insulation materials, on the basis of the torch screening test, developed by the Rock Island Arsenal Laboratory to date. In comparing the data for these two compounds to the data for similar compounds based on SBR (see Table V, F77 and F78), it appears that within the particular polymer-resin-asbestos combination in question, the use of NBR provides insulation materials superior to those in which SBR is the subber polymer. This apparent superiority of NBR over SBR may be due to the better resin compatibility of the former. Apparent differences between the thermal insulation properties of the SBR and NBR will be discussed at greater length later in this report.

Formula N141CF of Table VIII differs from N141F in that the rubber curatives were emitted, in an attempt to provide an undercure for the rubber portion of the compound, thereby increasing the ultimate elongation of the vulcanizate. The elongation did not increase but the torch performance became This is an important point. From the low elongation of compound N141F, it may be assumed that the matrix is predominantly plastic, rather than rubber, in nature. This is probably the reason why the presence or absence of rubbar curatives had little effect upon the stress-strain properties of the compound. Nevertheless, for some reason as yet not completely understood, the rubber curatives had an influence upon the performance of the compound in the torch test. Apparently the degree of crosslinking of the rubber, even though rubber is not the predominant ingredient, is important.

The last compound of Table VIII is the same as the first with the exception that it contains potassium exalate. This salt again proved to be detrimental, as shown by the lower index and higher erosion rate of the vulcanizate containing it.

In an attempt to utilize better the strength of fibrous materials, insulation specimens were prepared by forming laminates of asbestes cloth and rubber and of an organic heat resistant cloth (see code sheet) and rubber. In some cases, compounded rubber was sheeted out and placed between layers of the cloth. In other cases the compounded rubber was applied onto one side of the cloth by means of a two-roll calendar. In one instance, the rubber was dissolved in acetone, the cloth was scaked in the rubber solution and the coated cloth was air and vacuum dried. In all cases the plied-up layers were placed in a mold and cured under the same conditions as

TABLE VIII

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TORCH TEST DATA FOR FILLED VULCANIZATES BASED ON NBR

LALTS	Types and	Amounts (PHR	of Fillers	·			
RIA Formula No.	Phenol Furfural Resin	Phenol Long Furfural Fiber Potassium Resin Asbestos Oxalate	Potassium Oxalate	P200	H	Elong.	Tensile Strength, psi
N1.41.F	100	100	i	*17	2.5*	10	2400
N141F1	100	75	1	73	2.8	10	3410
N141F2	100	20	1	09	3.0	30	1880
N141CF	100	100	,	55	8.8	10	2880
N141F4	100	100	09	48	4.5	*	*
. <u></u>							

*Average of six tests.

^{**}This vulcanizate did not cure properly because of the presence of gas. Stress-strain properties were not determined.

were used for nonlaminated specimens. Table IX lists the major constituents, methods of fabrication and torch test data for the laminates.

None of the laminates of Table IX performed well in the screening test. Although P200 values are not available, it is estimated from the approximated specimen densities that the indices would range from 60 to 20. Sixty is a reasonably good index, however, all of the laminates delaminated during the torch test. The specimen in which the fabric was coated by being dipped in a solution of the rubber represented an attempt to provide good rubber to fabric adhesion. specimen, however, delaminated as readily as did the others. A comparison of the effectiveness of laminated and nonlaminated specimens can be made in only one instance. laminate containing nine layers of asbestos cloth is comparable in composition to the compound (S77C1F84, Table VII) containing 50 PHR long fiber asbestos. The time to 200°C for the nonlaminar compound was 43 and the ercsion rate was 6, therefore, it is obvious that the laminated compound showed poorer torch performance.

Experiments were conducted in an attempt to verify the postulate that the effectiveness of insulation which contains asbestos is related to the size of the asbestos particles or fibers. Table X summarizes data presented earlier in this report for SBR compounds each containing 100 PHR of a different particle or fiber size of asbestos, ranging from fine powder to long fibers. It is apparent from the data that the longest fiber asbestos provides the best insulation. data of Table XI further proves the superiority of long fiber over short fiber asbestos. All four entries in this table are for the same compound (S77ClF88, Table V) containing 100 PHR each of long fiber asbestes and phenol furfural resin. The compounds differ in the extent to which they were mill mixed. The first compound was mixed with the mill rolls as far apart as possible without losing continuity of the banded rubber. In this manner maximum asbestos dispersion with minimum fiber breakdown was achieved. After mixing, a pertien of the rubber was cured into torch specimens. The remainder of the rubber was then further milled by end over ending 10 times. A portion of this mix was removed and cursd. The third and fourth compounds received an additional 10 and 20 end over ends, respectively. The rubber which received the least mixing had long asbestcs fibers clearly visible on the surface of the cured specimens whereas the rubber which had received the most mill mixing had no visible fibers on the surface. The data of Table XI clearly show the superiority of the compound which had reseived the least mixing and thus centaining the lengest fibers. Both Tables X and XI show ____

TABLE IX

TORCH TEST DATA FOR RUBBER CLOTH LAMINATES

Ero- sion Rate, E	ភ	*	1	L	&	17
Time To Reach 2000C, seconds*	49	4 6	33	33	26	14
Fabrication Method	Cloth not calendared	.	Cloth soaked in rubber solution	Cloth cal- endared	t	2
Type of Cloth and No. of Layers	Organic heat resistant - 5	Organic heat resistant - 5	Open weave asbestos - 9	Open weave asbestos – 9	Organic heat resistant - 13	Organic heat resistant - 17
Major Constituents, PHR	Hydrated silica, 60	Hydrated silica, 60 - Phenol for- maldehyde resin #1, 120	Phenol fur- fural resin, 100	Ľ	A.	None (gum)
Polymer Type	65/35 butadiene/ acrylo- nitrile	8 .c. 1	76.5/23.5 butadiene/ styrene	E	2	9
RIA Formula No.	N139F2	N139F3	S77C1F43	S77C1F43	S77C1F43	S77CIF

*The performance indices could not be calculated because the densities of the laminates were not measured. Times to 200^{0} C are corrected for specimen thickness.

TABLE X

THE EFFECT OF ASBESTOS PARTICLE SIZE ON TORCH PERFORMANCE

RIA Formula No.	Approximate Particle or Fiber Size	P200	E.	Tensile Strength, psi
S77C1F22	0.18 - >0.5"	51	5	1050
S77C1F57	0.08 - 0.18"	18	14	800
S77C1F21	<0.08"	14	18	560
S77CF71	8 microns	10	26	540

TABLE XI

THE EFFECT OF DEGREE OF MILLING ON THE TORCH PERFORMANCE OF VULCANIZATES CONTAINING ASBESTOS

Extent of Milling*	P ₂₀₀	_E	Tensile Strength, psi
Minimum	59	3	2250
10 additional end over end	48	5	2280
20 additional end over end	44	5	1090
Maximum 30 additional end over end	34	7	Not measured

^{*}Compound used in this study was S77C1F88. See Table V for composition. All milling was performed with the ASTM standard roll gear ratio of 1.4:1. All results are the average of four tests.

good correlation among the variables of tensile strength, _ particle size (or extent of milling) and torch performance.

Although long fiber asbestos imparts good oxyacetylene torch resistance to rubber based vulcanizates, its rather high density (2.4 - 2.6) is a distinct disadvantage. In an effort to reduce the weight of asbestos-containing compounds, a study was made to determine the minimum amount of asbestos which would provide optimum torch resistance. The study was made with three polymers of the SBR and NBR types. A similar study was conducted with a hydrated silica filler (density 1.95). All compounds were mill mixed under the previously described conditions which optimize the retention of filler structure. Test data, shown in Table XII, clearly indicate that for each of the polymers investigated, there is a level of asbestos concentration which imparts optimum resistance to the screening test. Surprisingly, this level is approximately the same, 60 PHR, in each polymer. Asbestos loadings greater than 60 PHR do not provide better insulation. It is believed that the ratio of the concentrations of rubber and asbestos is very important to the quality of the insulation. certain proportion of asbestos is required to provide a thermally resistant char but a certain proportion of rubber is necessary to serve as a source of cocling gases.

The data of Table XII also show that 40 PHR of hydrated silica appears to be an optimum loading for the polymers investigated. The data further reveal a most interesting point, namely, that for the compounds which contain the optimum levels of asbestos or silica (60 and 40 PHR, respectively), those based on SBR are superior to those based on NBR. As noted previously, this difference could be due to the varying degrees of efficiency in which the polymers act as transpirational cooling agents. On the other hand, the difference could be due to the physical nature of the polymers. For example, the 55/45 butadiene/acrylonitrile polymer is much tougher, harder and more difficult to process than is the 76.5/23.5 butadiene/styrene polymer. It is quite possible that during the addition of asbestos to these two polymers on the mill, greater shearing forces are exerted on the asbestos in the case of the former polymer; resulting in an MBRasbestos compound containing shorter fibers than are present in the SBR-asbestos compound. It has already been shown that the shorter asbestos fibers lead to inferior insulation.

To study further the possible effect of polymer processability on the insulation effectiveness of vulcanizates containing asbestos, vulcanizates of nine different polymers, each compounded with 60 parts of long fiber asbestos, were prepared and tested in the torch. The solid polymers were

TABLE XII

THE RFFECT OF FILLER LEVEL ON TORCH PERFORMANCE

	è		3 2	Long	Long Fiber Asbestos, PHR	sbest	os, PHR			
	700		40		09	Ţ	S	0	100	
Polymer Type	P200	ল	P200	Ħ	P200	2	P200	×	P200	×
76.5/23.5 butadlene/ styrene*	45	7	46	. 60	1.7	ro.	45	2	51	5
65/35 butadiene/ acrylonitrile**	28	n	33	œ	80	∞′	25	7	35	7
55/45 butadienė/ acrylonitrile***	32	10	38	66	38	2	35	7	99	. 4
÷		Herden	Western Call Co.	- <u>;</u>	· •	-	×	-		

	1	Th ar	Sarated Silica, Pink	ICB.	Ĕ	ı	
270	20	_	4	0	09	_	
Polymer Type	P200	M	P200	Ħ	P200	×	
76.5/23.5 Bd/St*	14	22	37	8	23	13	
65/35 Bd/Ac**	22	15	33	10	25	11	
55/45 Bd/Ac***	20	16	25	10	25	11	
		•	. 5	-			

*Basic formulation - S77Cl **Basic formulation - N141C3D3 ***Basic formulation - N141C3 All results are the average of 4 tests.

mixed in the usual manner. The liquid polymers, their curatives and all but about two thirds of the asbestos, were mixed by hand stirring. The mix was too viscous at this point to permit the addition of the remainder of the asbestos by hand mixing, therefore, it was added by mill mixing. Even though the standard roll gear ratio of 1.4:1 was used, very little shesging action took place during the final mill mixing because the mill rolls were set about 1/2 inch apart and the compound was very soft. Table XIII gives the results of this work and also includes a qualitative evaluation of the processing characteristics of the polymers investigated. The data show excellent agreement between ease of processability and P200 and E values, thereby indicating that the soft, smooth, easily milled polymers, such as the liquid polymers and the first few solid polymers listed in Table XIII, probably do not cause asbestos fiber breakdown to the extent that the rougher, harder polymers do. Polybutadiene probably furnished poor insulation because it is a difficult polymer to process, not by virtue of its toughness but because of its weakness and tendency to crumble. Polymers which do not band well, but crumble on the rolls, usually must be milled with a tight nip, thereby increasing shearing forces which lead to ultimate fiber breakdown.

Table XIII also lists the behavior of each vulcanizate during burning in the screening test. Those compounds which had high performance indices and low erosion rates showed little or no evidence of loss of char or "spalling" during the torch test. Conversely, the poorer performing compounds spalled readily. Throughout this entire study, spalling has been noted to occur in those compounds in which the asbestos fibers are of short length. Here again is evidence that the integrity of the fiber must be retained in order to achieve optimum insulation. For example, those compounds based on the liquid SBR polymers and containing asbestos loadings as low as 40 PHR provide more efficient insulation materials than result from the use of 60 PHR asbestos with conventional solid SBR polymers. Undoubtedly the long fibers of asbestos suffer less breakage when mixed with liquid polymers than when milled with solid polymers.

In view of the good insulation materials developed from the liquid and the easy processing solid polymers in combination with 60 parts of long fiber asbestos, these polymers were further exploited by combining them with other types of fillers or lesser amounts of asbestos as noted in Table XIV. Some excellent materials resulted.

The second entry of Table XIV is one of special interest.

The polymer in this case is a blend of a butadiene/acrylonitrile (NBR) and polyvinyl chloride (PVC). The excellent

TABLE XIII

Slight spalling Slight spalling at beginning No spalling slight blowing Hvy. spalling throughout Hvy. spalling Hvy. spalling at beginning BEHAVIOR OF SPECIMEN ON BURNING at beginning No spalling No spalling No spalling throughout burning ON THE TORCH PERFORMANCE OF LONG FIBER ASBESTOS Excellent - Fairly smooth & soft band Very poor -low strength, Excellent - smooth, soft bands well POLYMER
PROCESSING
CHARACTERISTICS & tough, hands poorly & tough, bands poorly Excellent - Very scft, sticky band Excellent --Very soft, sticky band Poor - hard Poor - hard crumbles Good Good TENSILE STRENGTH, 1190 2450 1020 1800 2200 023 250 3330 1320 PRI THE EFFECT OF POLYMER PROCESSABILITY VUICANIZATES CONTAINING 60 PARTS KILONG., 195 80 85 25 75 8 5 80 8 Ä Ŋ 'n ۲ œ 40 m 4 P200 2 Տ 47 42 38 38 26 8 27 7500-12500 polses butadiene/acrylo-Blend of poly-vinyl chloride & Liquid SBR Viscosity: 500-2500 poises 65/35 butadione/ acrylonitrile 80/20 butadiene/ acrylomitrile 60/40 butadiene/ POLYMER TYPE 55/45 buta-diene/acrylo-nitrile Cis 1,4 poly-butadiene #2 76.5/23.5 butadlene/ styrene Liquid SBR Viscosity: styrene nitrile O. N141C3D3F13 S77ClD5F119 S77C4D2F119 S77C1D4F119 S77C4D3F119 N141C3D5F5 N141C3D4F5 S77CLF119 N141C3F5 RIA FORMULA

All results are the average of 4 tests

 \hat{E}_{i}

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	INSULATION MATERIALS	BASKD ON KASY	ROCES	SINC	PROCESS ING POLYMERS	88
RIA FORMULA NO.	POLYMER TYPE	TYPE AND ANOUNT (PHR) OF FILLER	P200	M	KLONG.,	TENSILE STRENGTH, P#1
S77C4DZF1	High viscosity liquid SER	Long fiber aspestos - 50	\$	n	9	2450
M41C305F19	MER, PVC blend	Long fiber asbestos - 40 Hydrated silica - 20	1.9	8	7.5	2200
STTC4DSF1	Low viscosity liquid SER	Long fiber asbestos - 50	Į.	ы	30	2130
S77C4D27118	High viscomity liquid SBR	Long fiber asbestos - 40	80	*	83 12 13	1660
N141C305F20	KBR, FVC blend	Long fiber asbestos - 60 Hydrated stlics - 40	88	•	55	3930
S77C4D3F118	Low viscosity Liquid SBR	Long fiber asbestos - 40	19	•	30	1170
N141C3D5F5	Mak, PVC blend	Long fiber asbestos - 60	8	<i>∌</i>	45	2450
S77C1F128	76.5/23.5 BBR	Long fiber asbestos - 40 Ultra fibe silics - 20	28	4	195	0\$6
S77C1F127	76.5/23.5 KBR	Long fiber asbestos - 40 Hydrated silica - 20	8	ю	160	01.6
S77C1F126	76.5/23.5 RER	Long filter asbestos - 40 Ultra fine carbon black- 20	2	4	220	006
877CLF119	76.5/23.5 BBR	Long fiber asbestos - 60	47	10	85	1190

torch performance of the vulcanizate based on this polymer may be due, in part, to the ease with which the long fiber asbestos is incorporated into the rabber, but may also be due to the presence of the PVC. It is known that polyvinyl fluoride, a polymer similar in many respects to PVC. volatilizes completely upon pyrolysis and does not form a char. Perhaps PVC acts similarly. If this is true, the blend of NBR and PVC would produce a larger volume of gases than would NBR alone, because undoubtedly NBR polymer produces some char rather than volatilizing completely. This increased amount of gas might produce a large enough occling effect to account for the superiority of the NBR/FVC blend. It was noted after the torah burning of the NBR/FVC dempound containing asbestos and silica (N141C3D5F19) that the specimen had blown and expanded. This behavior was not observed with any other similarly filled polymer compositions, thus indicating that the NBR/PVC blend did, perhaps, produce greater valumes of gas than did compounds based on other polymers.

The use of fine particle size carbon black or hydrated silica in conjunction with asbestos improved torch perfermance over that obtained with asbestos alone. Compounds N141C3D5F19 and N141C3D5F5 show this comparison for the compounds containing the NBR/PVC blend and the last four compounds of Table XIV show the comparison for vulcanizates based on a solid SBR. The reason for this apparent synergism is not understood.

Rubber manufacturers quite recently have been recommending the use of silicone rubber for rocket motor thermal
insulation. Examination of manufacturer's reports shows
that their recommendations are based on data obtained by
the use of low velocity, low heat flux torch systems. A very
limited study was made of insulation compounds based on methyl
vinyl and methyl phenyl vinyl silicone polymers. The torch
test results given in Table XV do not look precising, however,
the low viscosity, almost liquid character of many of the
silicone polymers, should make them very desirable as a
matrix for fibrous fillers. Further evaluation of liquid
silicone polymers is planned.

Several of the better insulation materials developed during the early portions of the work covered by this report were tested in attic rocket motor firings. Results of the firing tests conducted by the Atlantic Research Corporation are given in Table XVI. Results for the control compound used in these tests (a commercial rubber-based material) are also included. The insulations based on liquid polymers and on the NBR/PVC blend had not been developed at the time that the motor tests were conducted, however, the teach test data

TARI,R XV

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TORCH PERFORMANCE DATA FOR SILICONE VULCANIZATES

RIA FORMULA NO.	POLYMER TYPE	FILLER, PHR	P200	ELONG.,	TENSILE STRENGTH,
Z40F7	Methyl vinyl	Long fiber asbestos - 50	36	8 20	220
Z56C5F2	Methyl phenyl vinyl	Long fiber asbestos - 50	27	10	· · · · · · · · · · · · · · · · · · ·
Z56C5F	Methyl phenyl vinyl	None	23	15 810	1670
Z56C5F4	Methyl phenyl	Hydrated silica - 20	16	27 230	840
Z56C5F3	Methyl phenyl	Hydrated silica - 40	10	2 9	į

TABLE XVI

STATIC MOTOR FIRING AND TORCH TEST DATA FOR THE BETTER INSULATION MATERIALS

					ARC* MOTOR FIRING TRST DATA	R FIRING ATA	
BIA FORMULA NO.	POLYMER TYPE	F.11.2.RS. PIR	P200	N	CONVER- GENT, mils/sec.	PERIL PHERAL SLAB, mils/sec.	KLONG.
K1411	55/45 butadiene/ acrylonitrile	Phenol furfural - 100 Long fiber asbestos - 100	4	i m	1.6	4.8	10
Commercial Material	NBB	Phenolic regin Organic shit	. 77*	3**	3.1	5.0	m
S77C4D2FI	Liquid, high viscosity SBR	Long fiber asbestos - 50	99	og ³⁹⁷ og ³⁹⁷ , §	ı		Q
N141C3D5F19	NBR/IPVC Blend	Bydrated Killes - 20 Long filber albestos - 40	67	: M	0		75
STTC4D3F1	Liquid, low wiscomity SER	Long fiber separtos - 50	67	m			30
S77C1F88	76.5/23.5 butadiene/ styrene	Phenol furfural - 100 Long filler ambertos - 100	57	. 19	n d	æ K	ın
\$77C1F78	76.5/23.5 butadiene styrene	Metal oxides - 100 Long fiber asbestos - 100	23	4	. KJ	5.7	65

"Atlantic Besearch Corporation.

for such insulations are also given in the table. The results of the firing tests performed by the Allegany Ballistics Laboratory were classified and are not included in this report.

It is apparent from the data of Table XVI that the motor firing test results do not correlate in all cases with the results of the terch test. The convergent motor results show all three materials which were tested to be superior to the control. However, two of these three materials had poorer torch performance than that of the control. The peripheral slab motor test results would correlate with torch test data were it not for the one value of 3.8 which is out of line, Thus the convergent test rates as best the material with the highest torch performance, but the peripheral slab test rates as best one of the poorer torch performers. Similar lack of correlation between motor and torch test results have been observed by many investigators and is not really surprising, in view of the gross differences between the conditions of the tests, namely, exposure times, temperatures, pressures and environments.

The three compounds based on liquid SBR or the blend of PVC and NBR are listed in Table XVI merely to show the relationship of their torch test values to those of the compounds which were tested in motor firings. From this relationship it would seem reasonable to expect good performance of these three materials in the motor firing tests.

DISCUSSION

It was hoped that the exploratory work covered by this report would result in definite guidelines which would establish the types of materials needed to produce superior, rubber-based case insulation. One such guideline was established, namely, that effective insulation must contain materials which will form profuse amounts of hard, tenacious, erosion resistant char. It has been shown that gum elastomers do not, by themselves, form such chars, but that these same elastomers, when combined with certain types of fillers, produce char forming vulcanizates. The most effective fillers are: (1) the fibrous materials which combine good heat resistance with inherent strength, (2) resins which form highly crosslinked networks, and (3) reinforcing materials which increase the rubber crosslinked structure. Compatibility between filler and rubber is essential in all cases.

Several important observations have resulted from this study but proof of their general applicability is lacking. These observations are as follows:

- 1. The degree of crosslinking of the rubber component of the insulation is important to the net efficiency of the insulation. The degree of filler reinforcement and the type and amount of rubber curatives appear to be important in this respect.
- 2. Certain polymers seem to produce more effective insulation than others when each is combined with equal parts of the same fibrous filler. It is not known whether this is due to the ability of certain polymers to form more char or to produce more gas than others or whether the difference is due to the ability of the polymers to mix with the fillers without breaking down the fiber structure. To answer this question it will be necessary to evaluate a large number of liquid polymers of essentially equal viscosity, in order to eliminate the variable of fiber breakdown due to shearing forces between filler and polymer.
- 3. There appears to be an optimum ratio between filler and polymer for optimum torch performance. It is postulated that certain proportions of a char forming ingredient (filler) as well as a gas forming ingredient (rubber) are required. Verification of this postulate should be of great value to the future development of flexible insulations.

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